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Anisotropic creep modelling of the single crystal superalloy SRR99

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COVER: Equivalent stress in an embedded cell model with circular rigid fiber ($f = 0.5$) and hardening metal matrix ($E = 100$ GPa, $\epsilon_0 = 0.1\%$, $\sigma_0 = 100$ MPa, $N = 0.2$) at 3.8% total strain. This is Fig. 15a of the paper by M. Dong and S. Schmauder in this issue, p. 53.

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Anisotropic creep modelling of the single crystal superalloy SRR99

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Abstract

The starting point is a uniaxial non-linear rheological model, which performs the essential properties of the uniaxial behaviour under different loadings, such as monotonous and cyclic creep and LCF. A tensorial generalization of the uniaxial differential equations is derived by a projection method, which gives a general three-dimensional model obeying the crystal symmetries. An identification for the single crystal superalloy SRR99 at 760°C has been made for tensile creep tests in different orientations. The model can be used to determine the creep behaviour of structural elements such as bars, torsional shafts or turbine blades.

1. Introduction

The three-dimensional creep behaviour of crystalline materials can be extremely anisotropic. This fact has to be taken into account when modelling such materials. In the literature basically two approaches can be found for the three-dimensional modelling of the creep behaviour by constitutive laws. One approach is to use classes of slip systems and to introduce one-dimensional creep laws on the slip system level. By superposition of the shears in the slip systems a three-dimensional model is obtained which renders the symmetry properties of the corresponding crystal class. The range of applicability of such models, however, is limited to those creep mechanisms that are related to slip systems. Diffusional creep, for instance, is excluded, as are other creep mechanisms (see Refs. [1–3]).

The other approach, which is presently

favoured, uses algebraic tools such as tensor function representations and invariant theory. For the modelling of the one-dimensional behavior, a uniaxial rheological model is introduced which describes the behaviour under uniaxial creep tests in one orientation in its primary and secondary creep phase. Its governing differential equations have been generalized to three-dimensional anisotropic equations by means of a projection method which is described in Refs. [4,5]. We will limit our consideration to the primary and secondary creep phases, as little is known about the tertiary creep phase including anisotropic damage evolution.

The disadvantage of such a phenomenological approach, however, is that most of its material constants do not have an a priori physical meaning. The calibration of these constants has been carried out by comparing the experimental data with computer-simulated experiments. For

three-dimensional anisotropic constitutive models a large number of different experiments in different orientations is needed. In practice, however, tests in the high temperature regime are complicated and expensive. Thus the main difficulty of the present project was the calibration of the model by a rather limited amount of experimental data.

As the test data were obtained at a temperature of 760°C, the model has been calibrated for this single temperature and all material constants have to be considered as isothermal ones.

2. One-dimensional material model

The starting point of our constitutive modelling is the rheological four-parameter Burgers model (Fig. 1) which consists of two springs with elasticities C and K and two dampers with viscosities L and D (see Refs. [6,7]). All of these constants are assumed to be positive. The differential equations of the model are

$$\sigma^* = (C + K)\varepsilon^* - (C/L + C/D + K/L)\sigma + C/D\tau, \quad (1)$$

$$\tau^* = K(\varepsilon^* - \sigma/L), \quad (2)$$

where σ denotes the stress, ε the strain, and τ an internal variable which corresponds to the stress in the spring K . The strong non-linearity of the creep behaviour is described by the stress dependence of the viscosities

$$D = D_0 \exp(-B\sigma), \quad (3)$$

$$L = L_0 \exp(-B\sigma), \quad (4)$$

with the (positive) material constants D_0 , L_0 , and B . Thus, the viscosities remain constant during monotonic creep.

This one-dimensional model contains the five temperature-dependent parameters C , K , D_0 ,

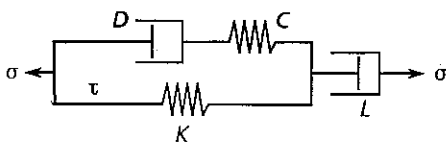


Fig. 1. Rheological Creep Model.

L_0 , and B , which have been calibrated by monotonic and cyclic creep tests of the single crystal SRR99 in [0 0 1] orientation. The model can easily be integrated numerically for given loading conditions. Experimental results and model calculations are presented in Refs. [8–11].

3. Three-dimensional material model

A three-dimensional generalization can be obtained by the projection technique described in Refs. [4,5] in analogy to the elastic behaviour. Hooke's linear elastic law for cubic materials is

$$\mathbf{S} = \mathbf{C}[\mathbf{E}] \quad (5)$$

where \mathbf{S} is the stress tensor, \mathbf{E} is the deformation tensor, \mathbf{C} is the fourth rank elasticity tensor with cubic symmetry, which can be generally represented as a linear combination in the following form:

$$\mathbf{C} = C_1\mathbf{P}_1 + C_2\mathbf{P}_2 + C_3\mathbf{P}_3 \quad (6)$$

of three independent elastic constants C_1 , C_2 , C_3 and the three structural tensors

$$\mathbf{P}_1 := 1/3 \left(\sum_{i=1}^3 \mathbf{e}_i \otimes \mathbf{e}^i \right) \otimes \left(\sum_{j=1}^3 \mathbf{e}^j \otimes \mathbf{e}_j \right), \quad (7)$$

$$\mathbf{P}_2 := \left(\sum_{i=1}^3 \mathbf{e}_i \otimes \mathbf{e}^i \otimes \mathbf{e}^i \otimes \mathbf{e}_i \right) - \mathbf{P}_1, \quad (8)$$

$$\mathbf{P}_3 := \left(\sum_{i,j=1}^3 \mathbf{e}_i \otimes \mathbf{e}^j \otimes \mathbf{e}^i \otimes \mathbf{e}_j \right) - \mathbf{P}_1 - \mathbf{P}_2, \quad (9)$$

where $\{\mathbf{e}_i\}$ is the lattice base, $\{\mathbf{e}^i\}$ its dual, and \otimes the tensor product. By applying the same procedure to the material constants in Eqs. (1), (2) we obtain the tensorial differential equations

$$\mathbf{S}^* = \mathbf{A}_1[\mathbf{E}^*] + \mathbf{A}_2[\mathbf{S}] + \mathbf{A}_3[\mathbf{T}], \quad (10)$$

$$\mathbf{T}^* = \mathbf{A}_4[\mathbf{E}^*] + \mathbf{A}_5[\mathbf{S}], \quad (11)$$

where \mathbf{T} is the tensor of internal variables, \mathbf{A}_i are material fourth rank tensors with cubic symmetry, which can be expressed as linear combinations of the three structural tensors \mathbf{P}_i :

$$\mathbf{A}_i = \alpha_{i1}\mathbf{P}_1 + \alpha_{i2}\mathbf{P}_2 + \alpha_{i3}\mathbf{P}_3, \quad (12)$$

with the scalar coefficients

$$\begin{aligned} \alpha_{1j} &= C_j + K_j, \\ \alpha_{2j} &= -(C_j/D_j + C_j/L_j + K_j/L_j), \\ \alpha_{3j} &= C_j/D_j, \\ \alpha_{4j} &= C_j/(C_j + K_j), \\ \alpha_{5j} &= -\alpha_{2j}\alpha_{4j}, \end{aligned} \tag{13}$$

where $C_j, D_j, K_j, L_j, j = 1, 2, 3$ take the part of elasticities and viscosities in different directions. It can be shown that these equations present the complete generalization of Eqs. (1), (2).

As before, the viscosities D_j and L_j depend on the applied stress intensity. In the one-dimensional case there is only one natural stress intensity, but in three dimensions we can define infinitely many. The following ansatz functions, which are based on the cubic integrity base, turned out to be sufficient for our model:

$$D_j = D_{0j} \exp(-\sum B_{ij}I_i), \tag{14}$$

$$L_j = L_{0j} \exp(-\sum B_{ij}I_i), \tag{15}$$

with the (positive) material constants $B_{ij}, i = 1, 2, 3, 4; j = 1, 2, 3$ and the following cubic scalar invariants (see Ref. [12]):

$$\begin{aligned} I_1 &= \sqrt{\sigma_{xx}\sigma_{yy} + \sigma_{yy}\sigma_{zz} + \sigma_{xx}\sigma_{zz}}, \\ I_2 &= \sigma_{xy}^2 + \sigma_{yz}^2 + \sigma_{xz}^2, \\ I_3 &= \sigma_{xy}\sigma_{yz}\sigma_{zx}, \\ I_4 &= \sigma_{xx}(\sigma_{xy}^2 + \sigma_{xz}^2) + \sigma_{yy}(\sigma_{yx}^2 + \sigma_{yz}^2) \\ &\quad + \sigma_{zz}(\sigma_{zx}^2 + \sigma_{zy}^2). \end{aligned} \tag{16}$$

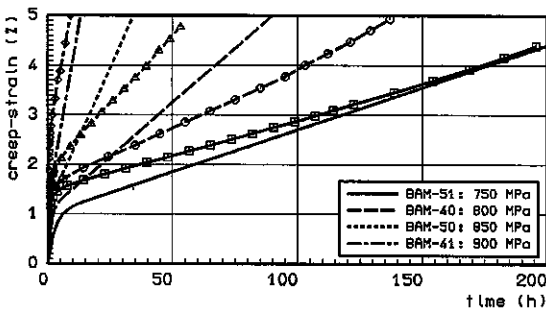


Fig. 2. Identification Tests in $\langle 001 \rangle$ -Orientation.

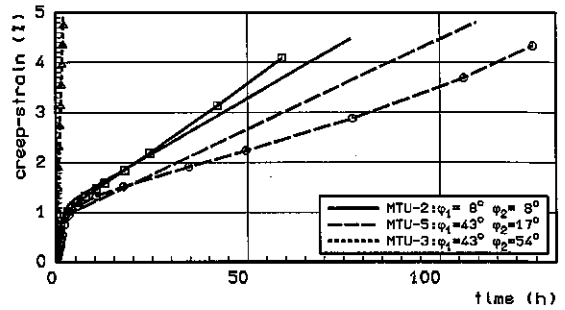


Fig. 3. Identification-Tests at 800 MPa.

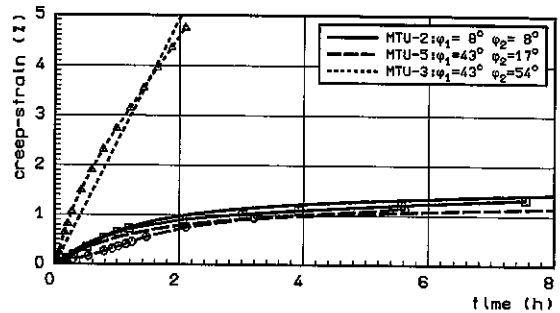


Fig. 4. Identification-Tests at 800 MPa.

The number of material constants can be reduced by the assumption that volume changes occur only elastically. Thus, $D_1^{-1} = L_1^{-1} = 0$ and $B_{i1} = 0, i = 1, \dots, 4$.

4. Material identification

The remaining material constants have been identified by tensile creep tests in different orientations for the superalloy SRR99 at 760°C. For that purpose, we divided the experimental data into two groups. The first (identification tests) was used to determine the material constants, the second for comparison of model and experimental behaviour (verification tests). In the diagrams, the symbols represent points of measurement, whereas the curves without symbols have been calculated by the model. φ_1 denotes the i th Eulerian angle, determining the orientation of the crystal relative to the specimen. $\varphi_2 = 0$ characterizes the $[001]$ orientation.

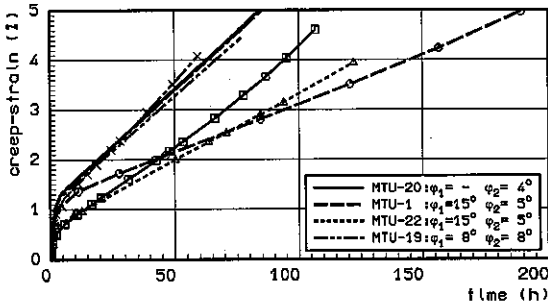


Fig. 5. Verification-Tests at 800 MPa.

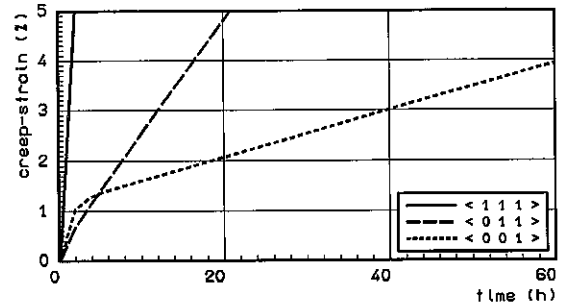


Fig. 7. Calculations at 800 MPa.

The identification tests consists of five tensile creep tests at different stress levels in $[0\ 0\ 1]$ -near orientation, performed at the author's institute (Fig. 2), and three tests with different orientations at 800 MPa, performed at Motoren- und Turbinenunion Muenchen (MTU) (Fig. 3, 4). The two diagrams show the same tests with different time-scalings. They demonstrate that the fit gives satisfactory results both in the primary and in the secondary creep phase.

The other tests were used as verification tests (Fig. 5, 6), all of them performed at MTU. It is a common feature of such creep tests that a large scatter band has to be expected. It can be clearly observed from Fig. 6 that specimens with similar orientations under equal conditions can produce rather different creep curves. Accordingly, the calibration of the model should not be overstressed. The agreement of the test and model data turns out to be sufficient if related to the scattering and the limited test data. The model is capable of reproducing the huge difference of the

creep strains in different orientations (see Fig. 3). In Fig. 7 the creep curves for the three extreme orientations has been calculated.

The three-dimensional model has been implemented to finite-element codes and used to study the spatial behaviour of torsional shafts with different orientations. Of course, a three-dimensional model requires justification and verification by tests under three-dimensional load conditions such as torsion and combined tension/torsion/pressure tests, which is done by a present project in the field. There are, however, principal problems to be solved for torsional tests on single crystals because of the loss of axial symmetry due to the anisotropy [13,14].

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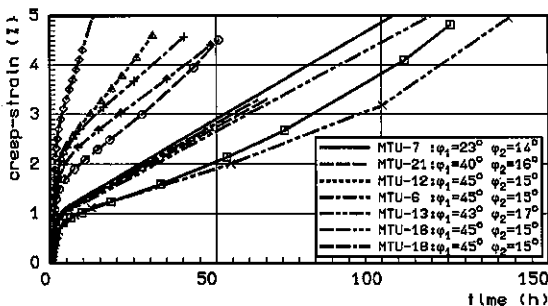


Fig. 6. Verification-Tests at 800 MPa.

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